

Stability of Aluminum in Low-Temperature Lithium-Ion Battery Electrolytes

Wishvender K. Behl and Edward J. Plichta

ARL-TR-1879 March 1999

19990420 102

Approved for public release; distribution unlimited.

The findings in this report are not to be construed as an official Department of the Army position unless so designated by other authorized documents.

Citation of manufacturer's or trade names does not constitute an official endorsement or approval of the use thereof.

Destroy this report when it is no longer needed. Do not return it to the originator.

Army Research Laboratory

Adelphi, MD 20783-1197

ARL-TR-1879

March 1999

Stability of Aluminum in Low-Temperature Lithium-Ion Battery Electrolytes

Wishvender K. Behl Sensors and Electron Devices Directorate, ARL

Edward J. Plichta

Communications-Electronics Command Research Development & Engineering Center

Approved for public release; distribution unlimited.

Abstract

We investigated the stability of aluminum at the high positive potentials encountered during the charging of lithium-ion cells. The electrolyte in these cells consists of solutions of lithium hexafluorophosphate and lithium methide in binary- and ternary-solvent mixtures of ethylene carbonate, dimethyl carbonate, and ethyl methyl carbonate. We performed the investigations with the controlled potential coulometry technique. We found that a protective surface film was formed on aluminum electrodes in these solutions and that this film protected the electrodes from further corrosion. The protective surface film was found to break down in lithium methide solutions at ~4.25 V versus a lithium reference electrode, and this resulted in increased corrosion of the aluminum electrodes at higher potentials. In contrast to lithium methide solutions, the protective surface film formed on aluminum electrodes in lithium hexafluorophosphate solutions was found to be quite stable and did not break down at potentials up to ~5 V.

Contents

1.	Introduction	1				
2.	Experimental Procedures	2				
3.	Results and Discussion	3				
4.	Conclusions	8				
Re	eferences	9				
Di	istribution	. 11				
Re	Report Documentation Page					
	Figures					
1.	Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L LiPF ₆ solution in 1:3 EC-EMC	5				
2.	Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L LiPF ₆ solution in 1:1:1 EC-DMC-EMC					
3.						
4 .	Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L lithium methide solution in 1:1:1 EC-DMC-EMC					
5.						
	Table					
1.	Freezing points of 1.0-mol/L solutions of LiPF ₆ in various solvent mixtures	3				

1. Introduction

The electrolytes used in commercial lithium-ion batteries freeze at approximately $-30\,^{\circ}\text{C}$ and thus their use is limited at lower temperatures. Recently, ethyl methyl carbonate (EMC) was identified [1] as a useful cosolvent for lithium-ion battery electrolytes. Since EMC freezes at $-55\,^{\circ}\text{C}$, it can be used as a co-solvent to extend the liquidus range of the lithium-ion battery electrolytes. We investigated a number of electrolyte solutions in binary and ternary mixtures of ethylene carbonate (EC), dimethyl carbonate (DMC), and EMC and recently proposed [2] a 1.0-molar solution of lithium hexafluorophosphate (LiPF₆) in EC-DMC-EMC (1:1:1 vol %) as the electrolyte for low-temperature applications of lithium-ion cells. We found the new electrolyte to have good conductivity and electrochemical stability. We also found that Li/LiCoO₂ and graphite/LiCoO₂ cells using the new electrolyte are operable at temperatures down to $-40\,^{\circ}\text{C}$.

The corrosion of aluminum-alloy current collectors for the positive electrodes in lithium-ion batteries has created considerable concern. In a recent paper [3], we reported our results on the stability of aluminum in lithium imide (lithium tris-(trifluoromethane-sulfonyl) imide, $\text{LiN}(\text{CF}_3\text{SO}_2)_2$) solutions. Aluminum was found to be unstable in these solutions at potentials above ~3.5 V versus a lithium reference electrode. The instability of aluminum in these solutions at high positive potentials was attributed to the breakdown of the protective surface film on aluminum. We also found that the surface film could be modified by using lithium tetrafluoroborate additive to prevent the corrosion of aluminum in lithium imide solutions at the high positive potentials encountered during the charging of lithium-ion cells.

This report summarizes our investigation of the stability of aluminum in several low-temperature electrolytes based on solutions of LiPF $_6$ or lithium methide (lithium tris-(trifluoromethane-sulfonyl) methide, LiC(CF $_3$ SO $_2$) $_3$) in binary and ternary solvent mixtures of EC, DMC, and EMC.

2. Experimental Procedures

We used LiPF₆ (Hashimoto, Japan) and lithium methide (Covalent Associates) as received. Ethylene carbonate, dimethyl carbonate, and ethyl methyl carbonate (all from Grant Chemicals) were dried over 4-Å molecular sieves before being used. Lithium foil (20-mil thick (Cypress-Foote Mineral Company)) packed over argon was opened in an argon-filled dry box (Vacuum Atmosphere Company) with a moisture content of less than 0.5 ppm.

We used a three-electrode system for all measurements. The reference and counter electrodes were both made by pressing lithium foil on a nickel screen. The working electrode consisted of a 1-mm-diam. aluminum wire that was heat-sealed in shrinkable Teflon[®] tubing. All potentials are referred to versus a lithium reference electrode.

We performed the controlled potential coulometry experiments with an EG&G Instruments, Inc., PAR (Princeton Applied Research), potentiostat/galvanostat (model 273). All experiments were computer-controlled using the EG&G PAR electrochemical analysis software (model 270). All experiments were performed inside a dry box.

3. Results and Discussion

We investigated the stability of aluminum in several LiPF $_6$ solutions in binary- and ternary-solvent mixtures of EC, DMC, and EMC. A number of these solutions had freezing points below ~-50 °C and thus were good candidates as electrolytes for low-temperature applications of lithium-ion cells. Table 1 shows the freezing points that we found for the various mixtures. The prefix numbers with each solvent mixture indicate the ratio of solvents by volume in each mixture, respectively. The aluminum electrode exhibited an initial potential of ~1.8 V versus the lithium reference electrode in these solutions, but increased to ~2.8 V and resulted in the formation of a surface film on the aluminum wire electrode. This film protects the aluminum substrates in lithium-ion batteries from further corrosion. The nature of the film on aluminum metal in these solutions has not been investigated but probably consists of aluminum fluoride or a species that contains fluoride.

Table 1. Freezing points of 1.0-mol/L solutions of LiPF₆ in various solvent mixtures.

Solvent mixture ratio (vol %)	Temperature (°C)		
1:3 EC-EMC	-65		
1:4 EC-EMC	-7 5		
1:1:1 EC-DC-EMC	-50		
1:1:2 EC-DMC-EMC	-65		
2:2:1 EC-EMC-DMC	-50		
1:1:3 EC-DMC-EMC	-65		

This study attempted to determine if the initial film formed on aluminum in these solutions was stable at the high positive potentials encountered during the charging of lithium-ion cells. We investigated the stability of aluminum at higher potentials by using the technique of controlled potential coulometry. The potential of the aluminum wire electrode dipped in the electrolyte was stepped up to a more positive value for 300 s, and the current response was plotted as a function of time. Typical plots obtained at potentials of 3.5 to 5.0 V in a 1.0-mol/L LiPF₆ solution in a binary 1:3 EC-EMC and a ternary 1:1:1 EC-DMC-EMC solvent mixture are presented in figures 1 and 2, respectively. Similar current density/time plots were obtained for aluminum electrodes in LiPF₆ solutions in other binary and ternary mixtures of EC, DMC, and EMC.

Figures 1 and 2 show that at each applied potential between 3.5 and 4.5 V, the current recorded at the aluminum electrodes quickly falls to a small steady-state value and shows no subsequent increase. Thus, the initial surface film formed on aluminum electrodes in LiPF $_6$ solutions appears to be quite stable and does not break down at potentials up to 4.5 V. At higher potentials, the current showed a slight increase after about 2 s. Since the LiPF $_6$ solutions in mixtures of EC, DMC, and EMC are known to undergo electrochemical oxidation [3] at potentials above ~4.5 V, the slight increase in current in the current density/time plots at 4.75 and 5.0 V may be regarded as due to the solvent oxidation. However, even at 4.75 V and 5.0 V, the steady-state currents were only slightly higher than

the steady-state currents at lower potentials. This indicates that the original protective film remains intact even at potentials above 4.5 V.

Recently, lithium methide solutions have been proposed [4–6] as thermally stable and highly conducting electrolytes for lithium-ion batteries. Therefore, we have also investigated the stability of aluminum in lithium methide solutions in several binary- and ternary-solvent mixtures of EC, DMC, and EMC. Typical current density/time plots obtained at aluminum electrodes at various potentials in 1.0-mol/L lithium methide solutions in a binary 1:3 EC-EMC and a ternary 1:1:1 EC-DMC-EMC solvent mixture are presented in figures 3 and 4, respectively. Similar plots were obtained in lithium methide solutions in other binary- and ternary-solvent mixtures of EC, DMC, and EMC.

The current density/time plots obtained at aluminum electrodes in lithium methide solutions were similar to those obtained in LiPF $_6$ solutions up to a potential of ~4.25 V only. At higher potentials, the current decreases initially but then begins to increase after ~100 ms. Also, the magnitude of the currents at potentials above ~4.25 V was much greater than that observed in LiPF $_6$ solutions. Therefore, it appears that the protective surface film initially formed on aluminum in lithium methide solutions breaks down at potentials above ~4.25 V. This results in high anodic currents due to the corrosion of the aluminum electrodes as well as the oxidation of solvents.

To compare the stability of aluminum in LiPF₆ and lithium methide solutions, we obtained the currents at various applied potentials from the current density/time presented in figures 1 through 4 at time t = 200 s and plotted in figure 5 as a function of the applied potential.

We see that the magnitude of the currents obtained at aluminum electrodes in LiPF $_6$ and lithium methide solutions is similar at potentials of up to ~4.25 V. At higher potentials, the currents in lithium methide solutions were much greater than those obtained in LiPF $_6$ solutions. Therefore, as shown before, it appears that in contrast to LiPF $_6$ solutions, the protective surface films formed on aluminum electrodes in lithium methide solutions are not stable at potentials above ~4.25 V.

Figure 1. Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L LiPF₆ solution in 1:3 EC-EMC.

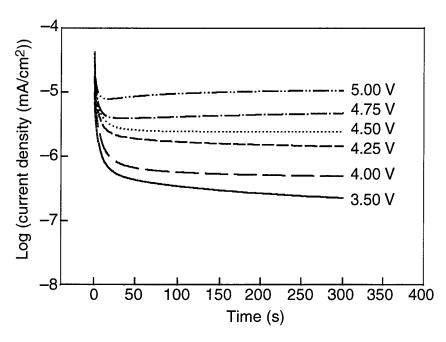


Figure 2. Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L LiPF₆ solution in 1:1:1 EC-DMC-EMC.

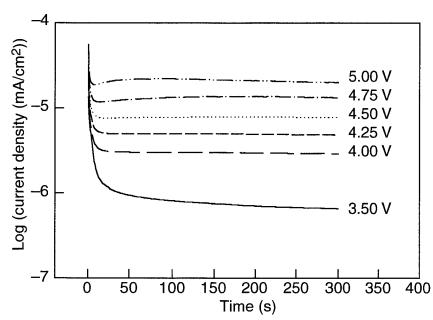


Figure 3. Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L lithium methide solution in 1:3 EC-EMC.

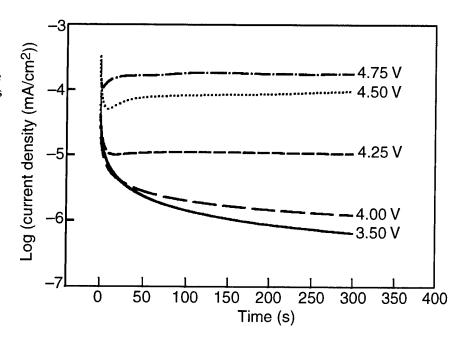


Figure 4. Current density/time plots obtained at aluminum electrode at various potentials in 1.0-mol/L lithium methide solution in 1:1:1 EC-DMC-EMC.

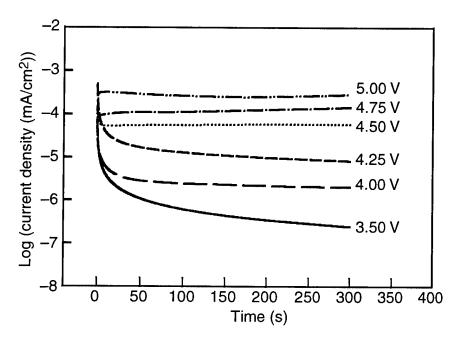
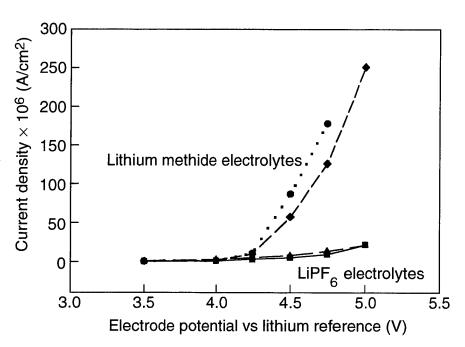


Figure 5. Current density vs applied potential plots at aluminum electrode in 1.0-mol LiPF₆ and lithium methide solutions in 1:3 EC-EMC (\blacksquare , \blacksquare) and 1:1:1 EC-DMC-EMC (\triangle , \blacksquare). Current densities were taken from current density/time plots presented in figures 1 through 4 at t=200 s.



4. Conclusions

Controlled potential coulometric experiments showed that the protective surface film formed on aluminum electrodes in lithium methide solutions in binary- and ternary-solvent mixtures of EC, DMC, and EMC breaks down at potentials above $\sim\!\!4.25$ V, and this results in increased corrosion at higher potentials. In contrast to lithium methide solutions, the protective surface film formed on aluminum electrodes in LiPF $_6$ solutions was found to be quite stable and did not break down at potentials up to $\sim\!\!5$ V.

References

- 1. Y. Ein-Eli, S. F. McDevitt, and R. Laura, J. Electrochem. Soc. 145, L1 (1998).
- 2. E. J. Plichta and W. K. Behl, *Proceedings of the 38th Power Sources Conference*, Cherry Hill, NJ (June 1998).
- 3. E. J. Plichta and W. K. Behl, J. Power Sources 72, 132 (1998).
- 4. L. A. Dominey, V. R. Koch, and T. J. Blakley, *Electrochim. Acta* 37, 1551 (1992).
- 5. F. Croce, A. D'Aprano, C. Nanjundiah, V. R. Koch, C. W. Walker, and M. Salomon, *J. Electrochem. Soc.* **143**, 154 (1996).
- 6. C. W. Walker, Jr., J. D. Cox, and M. Salomon, *J. Electrochem. Soc.* **143**, L80 (1996).

Distribution

Admnstr

Defns Techl Info Ctr Attn DTIC-OCP

8725 John J Kingman Rd Ste 0944

FT Belvoir VA 22060-6218

Ofc of the Dir Rsrch and Engrg

Attn R Menz

Pentagon Rm 3E1089

Washington DC 20301-3080

Ofc of the Secy of Defns Attn ODDRE (R&AT)

Attn ODDRE (R&AT) S Gontarek

The Pentagon

Washington DC 20301-3080

OSD

Attn OUSD(A&T)/ODDR&E(R) R J Trew

Washington DC 20301-7100

Advry Grp on Elect Devices

Attn Documents

Crystal Sq 4 1745 Jefferson Davis Hwy Ste 500

Arlington VA 22202

AMCOM MRDEC

Attn AMSMI-RD W C McCorkle Redstone Arsenal AL 35898-5240

CECOM

Attn PM GPS COLS Young FT Monmouth NJ 07703

CECOM Night Vsn/Elect Sensors Directrt

Attn AMSEL-RD-NV-D FT Belvoir VA 22060-5806

Commander CECOM R&D

Attn AMSEL-IM-BM-I-L-R Stinfo Ofc Attn AMSEL-IM-BM-I-L-R Techl Lib Attn AMSEL-RD-AS-BE R Hamlen

FT Monmouth NJ 07703-5703

Deputy for Sci & Techlgy

Attn Ofc Asst Sec Army (R&D)

Washington DC 30210

Dir ARL Battlefield Envir Dirctrt

Attn AMSRL-BE

White Sands Missile Range NM 88002-5501

Dir for MANPRINT

Ofc of the Deputy Chief of Staff for Prsnnl

Attn J Hiller

The Pentagon Rm 2C733

Washington DC 20301-0300

Hdqtrs

Attn DAMA-ARZ-D F D Verderame

Washington DC 20310

Hdqtrs Dept of the Army

Attn DAMO-FDT D Schmidt

400 Army Pentagon Rm 3C514 Washington DC 20301-0460

US Army Armament Rsrch Dev & Engrg Ctr

Attn AMSTA-AR-TD M Fisette

Bldg 1

Picatinny Arsenal NJ 07806-5000

Commander

US Army CECOM

Attn AMSEL-RD-CZ-PS-B M Brundage

FT Monmouth NJ 07703-5000

US Army CECOM Rsrch Dev & Engrg Ctr

Attn AMSEL-RD-AS-BE E Plichta

(15 copies)

FT Monmouth NJ 07703-5703

US Army Edgewood RDEC

Attn SCBRD-TD G Resnick

Aberdeen Proving Ground MD 21010-5423

US Army Info Sys Engrg Cmnd

Attn ASOB-OTD F Jenia

FT Huachuca AZ 85613-5300

US Army Natick RDEC

Acting Techl Dir

Attn SSCNC-T P Brandler

Natick MA 01760-5002

Director

US Army Rsrch Ofc

4300 S Miami Blvd

Research Triangle Park NC 27709

Distribution (cont'd)

US Army Rsrch Ofc Attn AMXRO-ICA B Mann PO Box 12211 Research Triangle Park NC 27709-2211

US Army Simulation, Train, & Instrmntn Cmnd Attn J Stahl 12350 Research Parkway Orlando FL 32826-3726

US Army Tank-Automtv Cmnd Rsrch, Dev, & Engrg Ctr Attn AMSTA-TA J Chapin Warren MI 48397-5000

US Army Train & Doctrine Cmnd Battle Lab Integration & Techl Dirctrt Attn ATCD-B J A Klevecz FT Monroe VA 23651-5850

US Military Academy Mathematical Sci Ctr of Excellence Attn MDN-A MAJ M D Phillips Dept of Mathematical Sci Thayer Hall West Point NY 10996-1786

Nav Rsrch lab Attn Code 2627 Washington DC 20375-5000

Nav Surface Warfare Ctr Attn Code B07 J Pennella 17320 Dahlgren Rd Bldg 1470 Rm 1101 Dahlgren VA 22448-5100

Marine Corps Liaison Ofc Attn AMSEL-LN-MC FT Monmouth NJ 07703-5033 USAF Rome Lab Tech Attn Corridor W Ste 262 RL SUL 26 Electr Pkwy Bldg 106 Griffiss AFB NY 13441-4514

DARPA Attn B Kaspar 3701 N Fairfax Dr Arlington VA 22203-1714

Hicks & Associates, Inc Attn G Singley III 1710 Goodrich Dr Ste 1300 McLean VA 22102

Palisades Inst for Rsrch Svc Inc Attn E Carr 1745 Jefferson Davis Hwy Ste 500 Arlington VA 22202-3402

Dir ARL Sensors, Signatures, Signal & Info prcsg Dirctrt (S3I) Attn AMSRL-SS Adelphi MD 20783-1197

US Army Rsrch Lab
Attn AMSRL-D R W Whalin
Attn AMSRL-DD J Rocchio
Attn AMSRL-CI-LL Techl Lib (3 copies)
Attn AMSRL-CS-AS Mail & Records Mgmt
Attn AMSRL-CS-EA-TP Techl Pub (3 copies)
Attn AMSRL-DC S Gilman
Attn AMSRL-SE J Mait
Attn AMSRL-SE-D E Scannell
Attn AMSRL-SE-DC W Behl (25 copies)
Adelphi MD 20783-1197

	JMENTATION PAG		Form Approved OMB No. 0704-0188				
Public reporting burden for this collection of information is gathering and maintaining the data needed, and complet collection of information, including suggestions for reduci Davis Highway, Suite 1204, Arlington, VA 22202-4302, a	s estimated to average 1 hour per response, incluing and reviewing the collection of information. Sing this burden, to Washington Headquarters Senand to the Office of Management and Budget, Pa	iding the time for reviewing end comments regarding to vices, Directorate for Infor perwork Reduction Project	ig instructions, searching existing data sources this burden estimate or any other aspect of this rmation Operations and Reports, 1215 Jefferso it (0704-0188), Washington, DC 20503.	s n			
1. AGENCY USE ONLY (Leave blank)	2. REPORT DATE March 1999	3. REPORT TYPE AN					
4. TITLE AND SUBTITLE Stability of Alum Battery Electrolytes 6. AUTHOR(S) Wishvender K. Behl (A (Communications-Electronics C Engineering Center)	5. FUNDING NUMBERS DA PR: AH94 PE: 62705A						
7. PERFORMING ORGANIZATION NAME(S) AND ADD U.S. Army Research Laboratory Attn: AMSRL-SE-DC 2800 Powder Mill Road Adelphi, MD 20783-1197			8. PERFORMING ORGANIZATION REPORT NUMBER ARL-TR-1879				
9. SPONSORING/MONITORING AGENCY NAME(S) AI U.S. Army Research Laboratory 2800 Powder Mill Road Adelphi, MD 20783-1197			10. SPONSORING/MONITORING AGENCY REPORT NUMBER				
11. SUPPLEMENTARY NOTES ARL PR: 9NE6V1 AMS code: 622705.H94							
12a. DISTRIBUTION/AVAILABILITY STATEMENT unlimited.	Approved for public release	; distribution	12b. DISTRIBUTION CODE				
We investigated the stability of aluminum at the high positive potentials encountered during the charging of lithium-ion cells. The electrolyte in these cells consists of solutions of lithium hexafluorophosphate and lithium methide in binary- and ternary-solvent mixtures of ethylene carbonate, dimethyl carbonate, and ethyl methyl carbonate. We performed the investigations with the controlled potential coulometry technique. We found that a protective surface film was formed on aluminum electrodes in these solutions and that this film protected the electrodes from further corrosion. The protective surface film was found to break down in lithium methide solutions at ~4.25 V versus a lithium reference electrode, and this resulted in increased corrosion of the aluminum electrodes at higher potentials. In contrast to lithium methide solutions, the protective surface film formed on aluminum electrodes in lithium hexafluorophosphate solutions was found to be quite stable and did not break down at potentials up to ~5 V.							
14. SUBJECT TERMS corrosion, methide hexafluoro	ethyl 15. NUMBER OF PAGES						
methyl	16. PRICE CODE						

17. SECURITY CLASSIFICATION OF REPORT

Unclassified

18. SECURITY CLASSIFICATION OF THIS PAGE

 ${\sf Unclassified}$

UL

20. LIMITATION OF ABSTRACT

19. SECURITY CLASSIFICATION OF ABSTRACT

Unclassified